

Formation of Icosahedral Phase in Bulk Glass Forming Ti-Zr-Be-Cu-Ni Alloy

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Formation of an icosahedral phase in the bulk glass forming $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ alloy during crystallization from amorphous phase and solidification from melt is investigated. The icosahedral phase with a size of 10 to 15 nm forms as a thermodynamically stable phase at intermediate temperature during the transformation from amorphous to crystalline phases such as Laves and β -(Ti-Zr) phases, indicating that the existence of the icosahedral cluster in the undercooled liquid. On the other hand, the icosahedral phase forms as a primary solidification phase even though the Laves phase is stable at high temperature, which is can be explained based on the high nucleation rate of icosahedral phase relative to that of competing crystalline Laves phase due to lower interfacial energy between icosahedral and liquid phases.

Key Words: Quasicrystal, Icosahedral, Metallic glass, Ti-Zr-Be-Cu-Ni, Crystallization

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INTRODUCTION

Precipitation of the icosahedral quasicrystalline phase from an amorphous matrix during annealing has recently attracted significant interest among scientists (Poon et al., 1985; Tsai et al., 1989; Holzer & Kelton, 1991; Lee et al., 2000; Xing et al., 2002), because of its potential to give insight about the relationship between amorphous and quasicrystalline structure and the growth kinetics of quasicrystals in undercooled liquid. Additionally the homogeneous distribution of the nano-scaled icosahedral quasicrystalline phase in crystalline or amorphous matrix results in significant improvements in mechanical properties (Xing et al., 1999; Kim et al., 2003a; Ren et al., 2011). However, all the icosahedral phases are formed as a primary phase during crystallization from the amorphous phase in these alloy systems. On the other hand, Al-based (Shechtman et al., 1984), Ti-Zr-Ni (Kim & Kelton, 1996) alloys form quasicrystals directly during rapid solidification from the melt. The formation of quasicrystalline

phases during solidification from the melt has been reported in the glass-forming Zr-based alloy systems (Saida et al., 2000; Park et al., 2005; Qiang et al., 2006; Mei et al., 2008). The formation of quasicrystalline phase in the alloy systems with a high glass forming ability that enables the formation of bulk amorphous alloy, indicate that the alloys can expand the practical applications as structural materials.

In this study we report the formation of icosahedral phase in the bulk glass forming $\text{Ti}_{40}\text{Zr}_{29}\text{Cu}_9\text{Ni}_8\text{Be}_{14}$ alloy during crystallization from amorphous phase and solidification from melt at low cooling rate.

MATERIALS AND METHODS

Alloys of nominal composition (at%) $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ was prepared by arc melting of high purity metals under an argon atmosphere. Rapidly solidified ribbon specimen was prepared under Ar atmosphere by melt spinning technique. The resulting ribbon exhibits thickness of about 30 μm and

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width of about 2 mm. The injection casting was performed to make bulk samples. Appropriate amounts of each alloy were remelted quartz tubes and injected through a nozzle into Cu mold, having cylindrical cavities of varying diameters from 4 to 6 mm and a height of around 50 mm. The structure of the ribbon and bulk sample was analyzed by X-ray diffractometer (XRD, RINT2200; Rigaku, Japan) with Cu $K\alpha$ radiation and conventional transmission electron microscopy (TEM, 2000FX; JEOL, Japan) and high-resolution electron microscopy (HREM, 4000EX; JEOL). The thermal property of samples was investigated by differential scanning calorimetry (DSC, DSC7; Perkin Elmer, USA) under Ar atmosphere at a heating rate of 0.67 K/s.

RESULTS AND DISCUSSION

Crystallization Behavior in $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ Alloy

Fig. 1 shows a DSC curve of the as melt-spun $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ alloy at a heating rate of 0.67 K/s. The DSC curve exhibits two exothermic peaks with onset temperatures of 643 and 766 K, respectively, and one high temperature endothermic peak with onset temperature of 829 K. A clear glass transition is not observed the alloy. The two exothermic reactions correspond to a two-step crystallization process and the high temperature endothermic reaction corresponds to a solid-state transformation of a crystallization product into high temperature phase.

In order to identify the crystallization product for each reaction, heat treatments were separately performed up to the finishing temperature of the each reaction for the alloy. Fig. 2 shows a XRD trace obtained from the as melt-spun and heat-treated the $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ ribbon alloy. The XRD trace from the as melt-spun specimen showed a broad diffraction peak, characteristic of the amorphous structure.

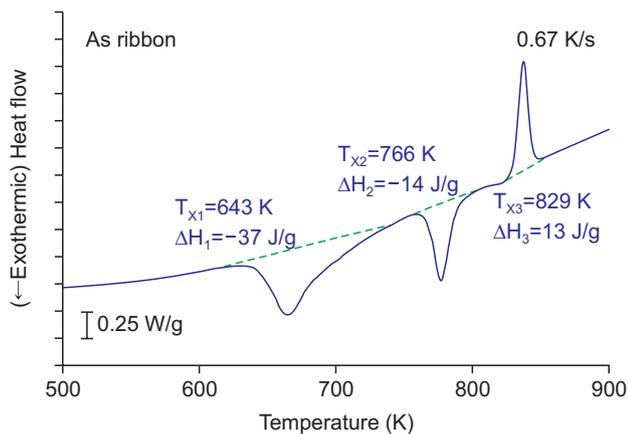


Fig. 1. Differential scanning calorimetry curve of the as melt-spun $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ ribbon during continuous heating at a heating rate of 0.67 K/s.

The specimen heated up to the first exothermic reaction ($T=763$ K) showed sharp diffraction peaks superimposed on a weak halo pattern, indicating a partial crystallization occurs during heat treatment. The diffraction peaks were analyzed to be an icosahedral phase with a quasilattice constant of $a_q=5.16$ Å. The XRD trace from the alloy heated up to second exothermic reaction ($T=813$ K) showed several sharp diffraction peaks from a mixture of an icosahedral phase and an MgZn_2 -type hexagonal Laves phase (C14). The XRD trace from the alloy heated up to 873 K, corresponding to the endothermic reaction, showed sharp diffraction peaks from a mixture of a hexagonal Laves phase ($a=5.23$ Å and $c=8.57$ Å) and a bcc solid solution β -(Ti-Zr) phase. The icosahedral phase transformed into a Laves phase through the endothermic reaction, indicate that the icosahedral phase is more thermodynamically stable at a lower temperature than

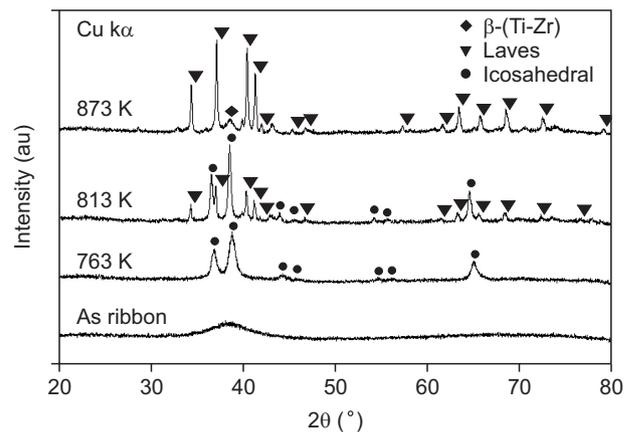


Fig. 2. X-ray diffractometer traces of the as melt-spun and heat-treated $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ alloy.

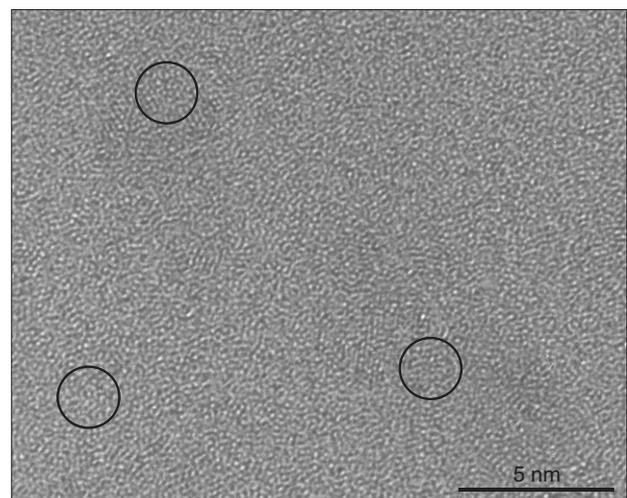


Fig. 3. High-resolution electron microscopy image of the as melt-spun $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ alloy.

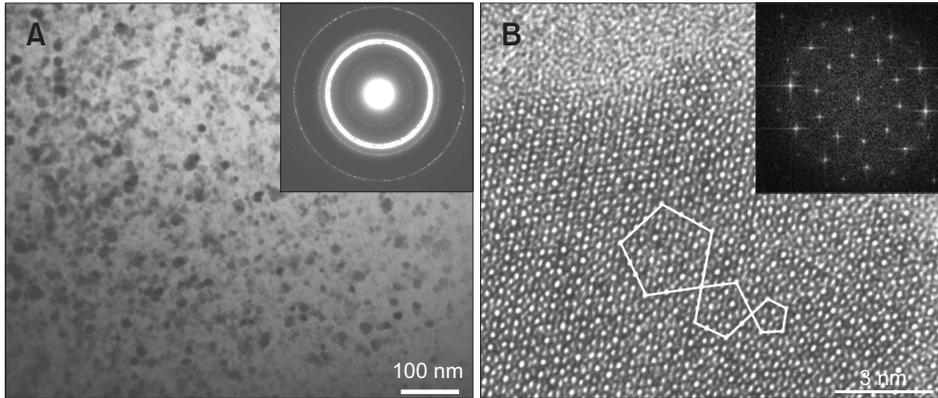


Fig. 4. Bright-field transmission electron microscopy image of the $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ ribbon alloy heated up to 763 K (A), self-aligned double patterning (inset in A), high-resolution electron microscopy image (B), and Fourier transformed pattern (inset in B).

the mixture of a Laves phase and a β -(Ti-Zr) phase. Recently, reversible transformation from the Laves phase and β -(Ti-Zr) phase to icosahedral phase has been reported in Ti-based alloy (Kim et al., 2003b).

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Fig. 3 shows a HREM image of the as melt-spun $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ alloy. The HREM image of the alloy shows a distribution of local ordered region as marked by circles, although the amorphous structure is confirmed by XRD and selected area diffraction pattern (SADP) results. A similar structure has been reported in the melt-spun Zr-based amorphous alloys (Xing et al., 2000; Saida et al., 2001), where the icosahedral phase is formed during the crystallization from amorphous phase. They suggest that ordered region is indicative of icosahedral cluster in the as melt-spun ribbon.

Fig. 4A show a bright-field TEM image and corresponding SADP from the $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ ribbon alloy heated up to 763 K. The SADP consists of several ring patterns superimposed on a diffuse halo patterns, indicating a mixture of crystalline and residual amorphous phases. All the ring patterns are analyzed to be an icosahedral phase. The alloy exhibits a random distribution of the icosahedral phase with a typical size of 10 to 15 nm in the amorphous matrix. Fig. 4B show a HREM image with the incident beam parallel to the five-fold axis and Fourier transformed pattern of the image taken from this alloy. The image showed a homogeneous distribution of sharp bright dots and their straight array on lines parallel to the five-fold directions as well as presence of some residual amorphous phase. The distribution of the bright dots form pentagons of various sizes associated with a scaling of the golden ratio τ , which can be interpreted as the three-dimensional Penrose lattice along the five-fold symmetry axis (Hiraga, 1997). Also, the HREM image showed a highly ordered structure of the icosahedral phase without any linear phason strain, which indicates the high structural quality of quasicrystal.

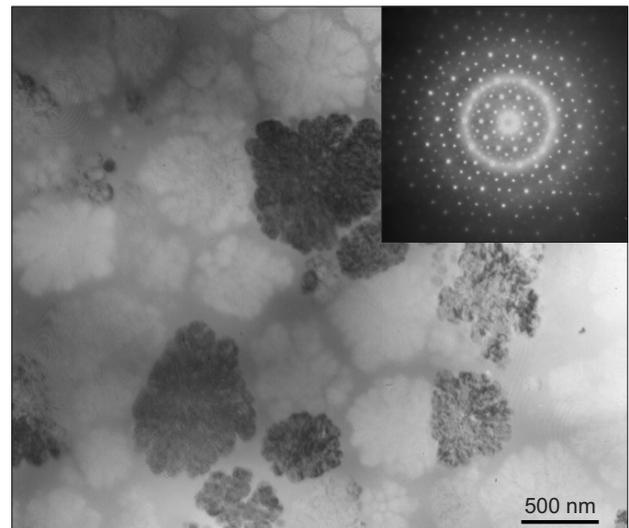


Fig. 5. Bright-field transmission electron microscopy image and self-aligned double patterning of the as injection-cast $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ bulk sample with diameter of 6 mm.

Fig. 5 shows a bright-field TEM image and corresponding SADP from the as injection-cast $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ bulk sample with diameter of 6 mm. The SADP showed the five-fold diffraction symmetry superimposed on a diffuse halo patterns, indicating a mixture of icosahedral and residual amorphous phases. Also, the pattern showed the sharp diffraction spots without diffuse scattering, indicating a reasonably ordered structure. The microstructure consisted of equiaxed dendrites of primary icosahedral phase with size of 300 to 500 nm and residual amorphous phase. TEM results indicate that in-situ composites consisting of stable icosahedral phase embedded in an amorphous matrix can be formed during solidification from the melt at moderate cooling rate. Above a critical cooling rate, fully amorphous bulk sample was fabricated by injection casting with diameter of 5 mm for the $\text{Ti}_{40}\text{Zr}_{29}\text{Be}_{14}\text{Cu}_9\text{Ni}_8$ alloy.

The formation of nano-sized icosahedral phase with large

volume fraction during the crystallization indicates that the nucleation rate of icosahedral phase is high, which is originated from the existence of the icosahedral cluster in the amorphous phase due to the high stability of the icosahedral short-range order during the solidification from the melt. The existence of stable icosahedral cluster can be understood by structural similarity in short-range order between liquid and the icosahedral phase. It has been reported that the interfacial energy between icosahedral phase and undercooled liquid is small, indicating the similar structure between icosahedral phase and undercooled liquid (Koster et al., 1997).

The local atomic structure of Laves phase is polytetrahedral type, corresponding to a large number of distorted tetrahedral configurations, which is similar to icosahedral configurations (Nelson & Spaepen, 1989). It has been reported that the solid-liquid interfacial energy decrease with increasing degree of polytetrahedral (or icosahedral) short-range order of the nucleating solid phase (Steinhardt et al., 1983; Holland-Moritz et al., 1998). It was estimated that the nucleation rate of icosahedral phase is slightly higher than that of polytetrahedral phase due to the small interfacial energy (Holland-Moritz, 1999). In this study, the formation of icosahedral phase as a primary phase during solidification even though the MgZn₂ type C14 Laves phase is stable phase at high temperature is

can be explained based on its high nucleation rate relative to that of Laves phase. Also, the formation of high ordered icosahedral phase during solidification is related to existence of icosahedral short-range order in undercooled liquid.

CONCLUSIONS

In the study, icosahedral phase was formed in the bulk glass forming Ti₄₀Zr₂₉Be₁₄Cu₉Ni₈ alloy during crystallization from amorphous phase and solidification from melt. The icosahedral phase forms as a thermodynamically stable phase at lower temperature during crystallization and transforms to Laves and β-(Ti-Zr) phases through the endothermic reaction at high temperature. The formation of nano-sized icosahedral phase during crystallization indicates the existence of an icosahedral short-range order in the undercooled liquid. The formation of icosahedral phase during solidification from the melt is related to high nucleation rate of icosahedral phase due to lower interfacial energy between icosahedral and liquid phases.

CONFLICT OF INTEREST

No potential conflict of interest relevant to this article was reported.

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